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# Exploring polycrystalline materials: high-throughput phase elucidation using serial rotation electron diffraction

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#### 12 Abstract

13 Rapidly and reliably elucidating the phases in polycrystalline materials is essential for developing new materials. Yet, crystals of many materials of biological, pharmaceutical, chemical, or industrial interest are too small (<1 14 μm) for routine X-ray diffraction (XRD) analysis. For complex materials, this can result in workflow bottlenecks 15 in high-throughput synthesis screenings favoured by industrial laboratories. With the increased prevalence of 16 electron diffraction as an alternative technique for materials characterization, we explore a series of zeolite 17 syntheses, resulting in typical polycrystalline products, via high-throughput phase identification using serial 18 19 rotation electron diffraction (SerialRED). Five zeolite phases were identified in one product, the most complex 20 mixture ever discovered in zeolite chemistry. Some of the phases are of ultra-low contents, similar unit cells, 21 and/or identical morphologies. Via automatically examining hundreds of crystals, SerialRED enables the reliable 22 and high-throughput phase analysis of products that XRD could not handle. It allows the exploration of more complex synthesis systems and provides new opportunities for rapidly developing novel polycrystalline materials, 23

24 greatly benefiting synthesis chemistry and material science.

## 25 Introduction

Crystalline materials demonstrate unique optical, electrical, thermal, mechanical, and/or magnetic properties and 26 27 have been extensively utilized. Nowadays, the synthesis conditions for exploring novel crystalline materials are becoming increasingly complex, motivated by growing and diverse demands. The resulting products are often 28 formed as polycrystalline products<sup>1-3</sup>, and therefore difficult to study using routine X-ray diffraction methods. To 29 unravel their phase compositions and atomic structures reliably at an early stage in the design of novel materials 30 aids in rationalizing the synthesis, evaluating properties, and steering the design for promising new 31 32 applications<sup>1,4-8</sup>. Meanwhile, recent trends of applying high-throughput synthesis screenings produce abundant polycrystalline products in a short time. Typical workflows require their phases to be identified and analysed at 33 commensurate time scales to avoid bottlenecks9-16, but for polycrystalline materials this has remained a 34 challenge<sup>17</sup>. 35

Since its first discovery over a century ago, X-ray diffraction has been well-established for phase analysis and 1 structure determination in chemistry. Single crystal X-ray diffraction (SCXRD) is the standard technique to 2 obtain accurate crystal structures, as long as crystals are large enough (about  $>5\times5\times5$  µm3 for a lab instrument) 3 4 to harvest enough diffracted signal. Meanwhile, this being a single-crystal technique, makes phase analysis challenging. For this purpose, powder X-ray diffraction (PXRD) is routinely used, but it comes with its own set 5 6 of limitations. Reflections with equal or similar *d*-spacings overlap in the one-dimensional pattern, making the phase and structure identification of polycrystalline materials by PXRD difficult and time-consuming, and 7 sometimes impossible<sup>17-19</sup>. Challenges arise when a polycrystalline product contains 1) multiple phases, 2) phases 8

9 with ultra-low contents (<1%), 3) phases with similar unit cells, and/or 4) structures with large unit cells or low

10 symmetries<sup>20</sup>. Some of the more interesting crystalline materials may therefore easily be ignored or overlooked.

11 Electrons, on the other hand, diffract much more strongly. Compared to X-rays, the amount of observed diffraction signal using electron radiation is about six orders of magnitude higher per unit of volume, which 12 enables useful data to be measured from crystals with sizes down to 50 nm<sup>20-24</sup>. Three-dimensional electron 13 diffraction (3D ED) is analogous to SCXRD, but operates at a much smaller scale. The development of 3D ED 14 15 methods has been rapid over the last decade, as researchers have discovered the benefits of this method to identify the phases and structures of polycrystalline materials that are too challenging to be studied by SCXRD/PXRD. 16 The structures of many different types of polycrystalline materials have now been determined using 3D ED<sup>25-32</sup>. 17 18 However, the searching of crystals for 3D ED data collection is still mostly a manual and time-consuming 19 endeavour. The selection of crystal for data collection is therefore subject to human bias and some of the phases 20 may be missed as a result. Although thousands of crystals are available on an electron microscopy grid, only a handful of crystals can be measured during a typical afternoon session. Therefore, we developed the high-21 throughput SerialRED method, which automatically screens crystals and collects 3D ED data, resulting in data 22 23 from hundreds of crystals in a product<sup>33,34</sup>. It combines the best of SCXRD (structure determination) with PXRD (phase analysis) in a single technique. Combined with cluster analysis, SerialRED enables objective, high-24 25 throughput phase analysis and structure determination for multi-phasic submicron-sized crystal products 26 (Scheme 1). This makes it an ideal technique to use in combination with the high-throughput synthesis screenings, helping us to accelerate the development of novel polycrystalline materials. 27

28 Here we demonstrate the application of SerialRED in the exploration of zeolites (Scheme 1). These are a class of typical metastable polycrystalline microporous materials that are widely applied in the industry<sup>35</sup>. Zeolites are 29 often naturally composed of submicron-sized crystals with complex structures and frequently synthesized as 30 31 multi-phases, which makes phase identification and structure analysis challenging<sup>17</sup>. In this work, we explore the 32 combination of multiple framework T atoms (T=Si, Ge, Al, and B) and a simple organic structure-directing agent 33 (OSDA) to synthesize novel zeolite materials. We initially characterized the phase composition of the resulting products by PXRD. Products that we could not identify with PXRD or manual 3D ED were then revealed by 34 SerialRED through high-throughput phase identification. This shows in practice the great advantage of 35 SerialRED in rapidly achieving reliable phase information where conventional methods fall short. 36



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Scheme 1 Life cycle for exploring complex polycrystalline zeolite products via high-throughput phase identification using SerialRED. A complex synthesis system consisting of multiple framework T atoms and a single, simple OSDA was designed to synthesize zeolites. The different framework T atoms were expected to trigger the formation of a diverse variety of structure building units and result in various framework structures where the OSDA mainly plays a pore-filling role. The phase composition was studied by SerialRED. The cycle is completed by using

7 the phase information in the rational development of novel zeolite materials.

#### 8 Results and discussion

9 Synthesis and phase diagram. In Table 1, we show the lifecycle for exploring the synthesis of novel zeolite 10 materials using two combinations of multiple framework T atoms ([Si,Ge,Al] and [Si,Ge,B]) and a cheap, simple, commercially available pore filling OSDA (4-dimethylaminopyridine, DMAP). The OSDA has previously been 11 used for the synthesis of zeolites with SFO, POS, and \*UOE type framworks<sup>36-38</sup>. Studies have shown that the 12 combination of framework T atoms directs the formation of specific structure building units<sup>39</sup>. Among them, Si 13 and Ge ([Si,Ge]) are mostly combined together to synthesize large or extra-large pore zeolites that are normally 14 unstable and lack active sites<sup>40</sup>. The addition of Al or B into the [Si,Ge] system usually triggers an increase in the 15 16 diversity of structure building units, introduce active sites, and result in thermal stable large or extra-large pore zeolites<sup>39</sup>. In our synthesis, the Si/Ge ratio was varied from 5 to 15 accompanied with the (Si+Ge)/T<sup>III</sup> (T<sup>III</sup>= Al 17 or B) ratios ranging from 0 to 100 to screen zeolite materials. Most synthesis batches give rise to crystalline 18 19 products, including five pure phases and a series of mixtures of bi- or multi-phases (Table 1, S1, and S2). In the [Si,Ge] system ((Si+Ge)T<sup>III</sup>=∞), two framework types, **TON** (1D, 10-ring) and **POS** (3D, 12×11×11-ring) were 20 obtained (Table 1, S1, and S2)<sup>41</sup>. With the gradual introduction of B, framework types NON (0D) and SFE (1D, 21

- 1 12-ring) were obtained (Table 1, S1, and S2), while the introduction of Al triggers the formation of NON and
- 2 **RTH** (2D,  $8 \times 8$ -ring) and a series of mixture products<sup>42-45</sup>.
- 3

OSDA/Si=0.6, HF/Si=0.6, H <sub>2</sub> O/Si=10				
		Si/Ge		
		15	10	5
$(Si+Ge)/T^{III} = \infty$		TON	TON+POS	POS
(Si+Ge)/Al	100	NON	Amor. +Den.+*UOE	Amor.+Den.+*UOE
	20	Amor.	Amor. +RTH+IWV+*CTH	RTH+*UOE+POS+IWV+*CTH
	15	Amor.+Den.+*UOE+IWV	Den.+RTH+*UOE+POS+IWV+*CTH	RTH+*UOE+POS+IWV+*CTH
	10	RTH	RTH+*UOE+POS+IWV+*CTH	RTH+IWV+*CTH
	5	Amor.	Amor.	RTH
(Si+Ge)/B	100	NON	Amor.	Amor.+POS
	20	Amor.+SFE	SFE	SFE+TON
	15	Amor.+SFE	SFE	SFE+TON
	10	SFE	SFE	SFE
	5	SFE	SFE	SFE

#### Table 1 Synthesis parameters and phase diagram<sup>a</sup>.

4 <sup>a</sup>T<sup>III</sup>=Al or B, Amor.=Amorphous, Den.=Dense phase (SiO<sub>2</sub>/GeO<sub>2</sub>). The phase information of mixture products that

5 are extremely difficult or impossible to be identified by PXRD are highlighted in the bold black frame and green

6 background.



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Figure 2 PXRD patterns (Cu Kα) and SEM images of (a) Product A and (b) B. Crystals with needle- and platelike morphologies are observed in Product A (Si/Ge=10, (Si+Ge)/Al=15). Uniform plate-like crystals are presented in
product B (Si/Ge=5, (Si+Ge)/Al=12.5). The crystals were severely smashed into smaller pieces for the SerialRED
experiments.

- The phase of these pure products, as well as most mixtures of bi-phases (with quite different unit cells or morphologies), were identified by PXRD with the assistance of scanning electron microscopy (SEM) (Figure S1 and S2). However, for some of the bi-phase products (with similar unit cells or morphologies) and most of the multi-phase products, only part of the phase information could be revealed by PXRD (Table 1). Figure 2a presents the PXRD pattern and SEM image of a typical complex mixture product (denoted as Product A). In Product A, only a significant **RTH** (needle-like) component was identified by PXRD (Figure S4a), but crystals with needleand plate-like morphologies were observed by SEM (Figure 2a). Therefore, **RTH** and an unknown phase with
- 19 plate-like morphology were regarded as the major phases in Product A in the beginning. To isolate the plate-like

crystals, we used a two-step heating crystallization program in the synthesis to avoid the formation of **RTH** (see the synthesis details in supporting information). The obtained purified product (denoted as Product B) shows very high crystallinity and uniform plate-like crystals (Figure 2b and S5). Although the SEM images indicated that Product B is very likely to be a pure phase, we were unable to index its PXRD pattern using the SVD-index method implemented in *Topas V*6<sup>46</sup>.

6 Therefore, we turned to manual 3D ED, a technique that enables us to collect high-quality electron diffraction 7 data for an isolated submicron-sized single-crystal. This revealed a zeolite **IWV** (2D, 12×12-ring, plate-like 8 crystal) in Product B (Figure S6)<sup>5,47</sup>. Subsequently, **IWV** is also identified in Product A by comparing the PXRD 9 patterns of Product A and B (Figure 2 and S4b). However, the PXRD pattern of Product B could only be indexed 10 partially with the unit cell of **IWV** (Figure S7). When we tried to fit the PXRD pattern of Product A with two 11 phases, **RTH** and **IWV**, we were also left with unindexed peaks (Figure S4b). Therefore, some of the phases in 12 these two products may still be ignored by PXRD characterization and even by manual 3D ED study.

13 High-throughput phase identification using SerialRED. To systematically approach the problem and reveal all the phases in Product A and B, we used the SerialRED method that we developed in our lab<sup>33</sup>. SerialRED 14 15 enables us to collect 3D ED data from hundreds of individual crystals automatically, with the aim to perform high-throughput phase analysis and structure determination. The SerialRED experiments were performed using 16 the protocol implemented in the software Instamatic and on a trace amount of sample<sup>33,34</sup>. For Product A, the 17 18 SerialRED routine ran for 6 hours resulting in 321 3D ED data sets (Figure S8). On our setup, we have integrated the program *DIALS* for on-the-fly unit cell identification for every data set<sup>24</sup>. As a result, 146 data sets were 19 indexed with the corresponding unit cell parameters. Out of these, 77 out of 146 indexed data sets with a rotation 20 range larger than 20° were used for phase analysis. The high-throughput phase analysis was performed by feeding 21 the identified unit cells into the hierarchical clustering analysis (HCA) algorithm implemented in the package 22 edtools (Figure S9)<sup>33,48</sup>. The euclidean distance between the unit cell parameters was used as the metric for 23 clustering. 24

25 Figure 3 and Table S1 present the clustering and phase analysis results of Product A. The cluster analysis revealed 26 three more zeolite framework types, \*UOE (2D, 10×8-ring, needle-like), POS (needle-like), and \*CTH (2D, 14×12-ring, plate-like), in addition to **RTH** (needle-like) and **IWV** (plate-like) (Table S1, Table S2)<sup>49</sup> that were 27 already identified. RTH, IWV, and \*CTH would be the major phases as indicated by the clustering results. The 28 29 data sets in each cluster were merged, and the framework structures could be determined directly by SHELXT (Figure 3, Figure S10 and Table S3)<sup>33,50</sup>. These findings are corroborated by the Pawley fit of PXRD pattern of 30 all phases using the routine implemented in the program TOPAS V6<sup>46</sup>. The PXRD pattern of Product A can be 31 32 well fitted with phases RTH, IWV, and \*CTH, which confirms that they are the major phases in Product A (Figure S4c). For \*UOE and POS, which have morphologies almost identical to RTH (Figure 2), no obvious 33 reflections from them were observed, and no significant improvement of the Pawley fit was obtained after 34 including them (Figure S4c and S4d). This clearly highlights the great advantage of SerialRED in picking up 35 minor phases that could not be detected by PXRD and/or SEM. 36



Figure 3 Dendrogram showing the results of the HCA of Product A. The y-axis is the euclidean distance between
the unit cell parameters and is described in supporting information. HCA revealed five zeolite phases including RTH,
IWV, \*CTH, \*UOE, and POS by setting the cut threshold at 2.0. Among them, RTH, IWV, and \*CTH are the major
phases. The unclassified data sets (marked by pink stars) could not be identified. These could be data from crystal
agglomerations or of otherwise bad quality, which both result in inaccurate unit cell parameters with large deviations
(Figure S11). The unit cell parameters of all data sets are shown in Table S1.

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8 In Product B, the phase **\*CTH**, which has a unit cell, morphology, and crystal size very similar to those of **IWV**, 9 was also identified by SerialRED (Figure S12, S13, S14 and Table S4). Notably, we missed it in our initial assessment of the sample by using manual 3D ED, which can be attributed to a combination of small sampling 10 size and crystal selection bias. The larger number of crystals sampled by SerialRED gives us a better statistical 11 overview and thus more objective phase information of Product B. HCA of the SerialRED data revealed 75% 12 IWV : 25% \*CTH from 36 crystals (counted based on the number of detected crystals, Table S4). Profile 13 14 refinement of Product B against synchrotron powder X-ray diffraction (SPXRD) data using TOPAS V6<sup>51</sup> (Figure S15, Table S5) gave a phase composition of 67% IWV : 33% \*CTH. The metrics differ in that SPXRD profile 15 analysis determines weight percent and cluster analysis counts the number of crystals. Although our sampling 16 17 size is still relatively small, this indicates that SerialRED is capable of quantitative phase analysis for crystalline materials as suggested previously<sup>33,52</sup>, as long as the number of crystals collected is large enough. 18

New opportunities provided by SerialRED for developing zeolite materials. With all the phases revealed by SerialRED, the roles of different framework T atoms Si, Ge, Al, and B in our synthesis system become clear. We noticed that the framework structures of NON, TON, SFE, IWV, and \*CTH synthesized using the same OSDA (DMAP) are all highly related and containing similar chains (*ton-, non-, IWV-,* and *CTH*-chains) and layers (*ton-, <i>non-, IWV-,* and *CTH*-layers) consisting of similar or identical building units (*ton* or  $5^26^2$ , Figure S16 and 4). Based on the common features in their framework structures and synthesis conditions, the formation of those chains, layers, or building units can be attributed to a Si-based system with a small amount of Ge (Figure S16

1 and 4). Meanwhile, the differences in the framework structures and synthesis conditions show that the formation 2 of non and d4r units were triggered by a small amount of B or Al and a considerable amount of Ge, respectively. Therefore, by introducing considerable Al and Ge into the synthesis system of TON, IWV and \*CTH (both 3 containing ton, non, and d4r units) were formed (Figure 4). The close structural relationship between IWV and 4 \*CTH also explains the difficulty in synthesizing each in pure form. For SFE, its framework structure relates to 5 6 that of **TON** through  $\sigma$ -expansion (Figure S17), which we achieved by introducing significant B into the synthesis system of **TON**. This clearly shows the role of B in promoting the formation of small *s*4*r* units<sup>39</sup>. In our designed 7 synthesis system, zeolite structures SFE, IWV, and \*CTH, which are prepared using bulky and expensive 8 OSDAs (Table S6), were successfully synthesized using DMAP as OSDA. DMAP is simple, commercially 9 available, and therefore economically more viable for the large-scale production of SFE, IWV, and \*CTH. In 10 addition, our phase analysis results show that the two new zeolites \*UOE and \*CTH reported recently had 11 already been synthesized in our system six years ago<sup>38,49,53</sup>. Were we able to identify these phases as novel at that 12 time, we would have optimized our synthesis for those frameworks. Yet, we discarded the product as too complex 13 to analyze. Therefore, we believe that SerialRED has a lasting place in our synthesis pipeline to identify 14 interesting materials at an early stage of the synthesis. This approach is compatible with the way favoured by 15 large-scale applications in industry, using SerialRED as a powerful and automated screening method for the 16 17 exploration of such complex synthesis systems.



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Figure 4 Structural relationship between IWV and \*CTH (only one type of ordered frameworks was presented here). The frameworks of them are highly related and can both be constructed by *ton*, *non*, and *d4r* building units in a very similar manner. The very similar *IWV*-layers and *CTH*-layers, which are also closely related to those of **TON** and **NON** (Figure S16), are built from *ton* units. The incorporation of *non* units on the *IWV*-layers and *CTH*-layers results in *IWV\_1*-layers and *CTH\_1*-layers, respectively. The neighbouring *IWV\_1*-layers and *CTH\_1*-layers are then both

connected by sharing *d*4*r* units and forming **IWV** and **\*CTH** frameworks, respectively.

2 B. The catalysis testing of product B was conducted on the isomerization of bulky isopropylnaphthalene molecules, which is a reaction to prepare important monomer 2,6-diisopropylnaphthalene for the advanced 3 polyester fibers, films, and plastics (Figure S18-22 and Table S7, S8) <sup>54,55</sup>. Its performance was compared with 4 those of MOR (2D, 12×12-ring) and SFE zeolites, which are excellent catalysts for the isomerization of 5 isopropylnaphthalene<sup>55-58</sup>. Our results show that Product B with moderate acid properties has much higher 6 catalytic efficiency than those of MOR and SFE (Figure S21, S22, and Table S8). This indicates that Product B 7 8 has more accessible active sites for isopropylnaphthalene than those of MOR and SFE. This is because Product B contains 2D large-pore (IWV, 12×12-ring) and extra-large-pore (\*CTH, 14×12-ring) zeolites, which have 9 10 advantages in diffusing bulkier molecules. These offer us better ideas to develop the application of this catalyst

Catalysis. In addition, the phase information also helps the understanding of the catalytic properties of Product

11 in industrial processes that involve bulky molecules such as oil refining and fine chemical synthesis.

#### 12 Conclusions

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13 In this work, using the exploration of zeolite materials as a practical example, we demonstrated the great benefit of SerialRED in developing polycrystalline materials. Being able to automatically screen a large number of single 14 15 crystals and collect 3D ED data, SerialRED offers new opportunities for rapidly accessing the reliable phase information of complex polycrystalline products via high-throughput screening. Five zeolites RTH, \*UOE, POS, 16 IWV, and \*CTH, some of which were even with extra-low content, similar unit cells, and/or similar 17 18 morphologies that were unable to be detected/identified by PXRD or even manual 3D ED, were revealed in the 19 most complex zeolite mixture by SerialRED. The phase information helps to understand the roles of different framework T atoms and to accelerate the development of zeolite materials by being able to identify interesting 20 phases at an early stage in the synthesis development. We also presented the possibility of the quantitative phase 21 22 analysis using SerialRED data that showed general agreement with PXRD data. In addition, the SerialRED experiments are performed on a trace amount of sample, which is desirable for the nanomole-scale high-23 throughput synthesis chemistry. These advantages of SerialRED would essentially expand the scope of synthesis 24 25 chemistry for interesting polycrystalline materials. Undoubtedly, apart from zeolite materials, SerialRED can 26 also be a promising method with many future opportunities for facilitating the exploration of mineral, metal/metal oxide, ceramic, semiconductor, and also polymorphism in organic crystals including drugs, etc. 27

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## 6 Author contributions

7 X. Z. directed the SerialRED study. W. Y. directed the synthesis of zeolites. Y. L. designed and performed the

8 synthesis experiments of zeolites. B. W. and Y. L. performed the SerialRED data collection and analysis. B. W.

9 and S. S. developed the SerialRED method. B. W. and Y. L. improved the data processing of SerialRED. B. W.

10 conducted the structure refinement against the SerialRED data. Y. L. performed the Rietveld refinement, Pawley

11 fit, catalytic study, and all other characterizations. J. S. supported the manual 3D ED experiments on the phase

12 identification of IWV in product B. Y. L., B. W., and S. S. wrote the initial draft. All authors reviewed and

13 commented on the manuscript.

## 14 Competing interests

15 The authors declare no competing interests.